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 $\sim$ 343.7F/g at 15 mV/s. As a result, electrodes based on produced MnO2/rGO composites

have the potential to exhibit excellent capacitive properties.

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# **Development of MnO2/rGO Composite Electrodes for Enhanced Electrochemical Supercapacitor Performance**

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# **1. INTRODUCTION**

Supercapacitors differ from conventional energy storage technologies like batteries and fuel cells due to their high-power density, enormous energy storage capacity, and extended cycle life. When compared to conventional capacitors, electrochemical supercapacitors offer superior capacitance values and higher energy density per unit area[1]. Supercapacitors are categorized into two primary varieties based on their energy storage methods: pseudo-capacitors and electric double-layer capacitors (EDLCs). To expedite electrosorption, intercalation between electrodes, and redox processes, pseudocapacitors and EDLCs are currently necessary[2], [3]. Pseudo-capacitors have recently gained significant interest as a potential energy storage device, as the primary charge transfer at the electrode-electrolyte interface occurs as a result of the flow of a faradic current[4]. This process is known as electrosorption or the faradaic reaction. Electrodes in pseudo-capacitors are typically made of conjugated conducting polymers, transition metal sulfides, or transition metal oxides. When compared to the materials used alone, the combination of metal oxides and carbon-based materials shows much better charging/discharging ability, capacitance value, and cyclic stability[5], [6].

aerogels because of their strong structure and high adsorption capacity[7]. Electrode materials made of manganese dioxide  $(MnO<sub>2</sub>)$  are now the most interesting option for electrochemical energy storage systems due to its capacitive responsiveness and cycling characteristics. The comparatively high surface area, which includes a high pore volume and size, of  $MnO<sub>2</sub>$ has made it a highly sought-after catalyst for use in a wide range of catalytic processes[8], [9]. The electrical conductivity of  $MnO<sub>2</sub>$ is typically low, but it can be enhanced by combining it with carbon compounds. One of the best ways to increase the capacitance and rate capabilities of supercapacitors is to create a  $MnO<sub>2</sub>$  composite<sup>[10]</sup>. Electrode materials in a wide range of electrical devices often make use of graphene derivatives such as GO, rGO, etc., due to their huge specific surfaces, excellent electrical conductivity, and ease of mass manufacture[11]. rGO's hydrophilicity and presence of charged structures give it excellent dispersibility in both water and organic solvents. Also, supercapacitors can't use pure  $MnO<sub>2</sub>$  since the material has a high conduction resistance and a very low capacitance value[12], [13]. An appealing alternative electrode material for electrochemical supercapacitors is a combination of graphene oxide and manganese dioxide.

A wide variety of possible catalytic uses have been shown by

Due to their low electronic conductivity, prior research has shown that the specific Cs of supercapacitors based on metal oxides decreases substantially as the scan rate increases[14], [15]. Adding metal oxides to carbon-based materials makes sense since composites take advantage of both components' strengths while mitigating their weaknesses. Since the flaws of graphene and metal oxides, especially  $MnO<sub>2</sub>$ , can be balanced out by their combination, the electrochemical performance of the two materials can be enhanced $[16]$ . As an example, rGO-MnO<sub>2</sub> composites were synthesized by authors [17] using a hydrothermal approach; with a sweep rate of  $0.5 \text{ Ag}^{-1}$ , they achieved a specific capacitance of 300 F g1. Authors [18] produced MnO2 nanotubes/RGO and showed that at a rate of 0.1 Ag-1, the material had a discharge capacity of 263 mAhg1. Because of their unique physicochemical characteristics, TiO<sub>2</sub> particles—and MnO<sub>2</sub> in particular—have attracted a lot of attention. Many different things have made use of these particles: gas sensors, catalysts, support materials, pigments, and even environmental purification systems [19], [20]. It is thought that high performance electrochemical devices could benefit from electrodes made of a combination of  $MnO<sub>2</sub>$  and GO/rGO.

Combining  $rGO$  with  $MnO<sub>2</sub>$  to create an effective composite may offer a significant surface area for intercalation and deintercalation, as demonstrated in this study. Electrochemical supercapacitors made of a sol-gel synthesized  $MnO<sub>2</sub>$  and rGO composite have been developed utilizing an inexpensive technique. The electrochemical study shows that the constructed MnO<sub>2</sub>/rGO composite electrode has created a channel for ions to easily go through, which could lead to a decrease in ionic resistance.

#### **2.MATERIALS AND METHODS**

MnO<sub>2</sub> was produced using an easy solvothermal method. A mixture of 12 g of titanium (IV) butoxide, 40 mL of acetone, and 1.08 mL of acetylacetone was stirred continuously at first. Next, 15.2 mL of acetone and 2.4 mL of water were delicately combined to create a sol that subsequently turned into a wet-gel. The new solution was gradually mixed with the old one. The gel was allowed to grow at room temperature for 24 hours before being placed in a 100 mL autoclave lined with Teflon. Then, MnO<sub>2</sub> was produced by programming the autoclave to run for four hours at a temperature of 140ºC. The final result of the solvothermal procedure was filtered and then cooked in a prepared oven overnight at 60ºC. The last step was to gather all of the ingredients and transfer them to a vial for later use.

After synthesizing a series of  $MnO<sub>2</sub>/rGO$  nanocomposite, the process of precipitation was employed. Sonicating a mixture of 10 mL of ethanol with a specific amount of as-synthesized  $MnO<sub>2</sub>$ and rGO for 2 hours at room temperature is a common procedure in most experiments. Filtration was subsequently employed to obtain the MnO2/rGO nanocomposite. After filtering, the specimen was dried in an oven at 60ºC for the night. Then, it was ground with an agate mortar. Changing the rGO weight ratio allowed for the synthesis of three separate nanocomposites. Each sample is called a  $MnO_2/1GO$ ,  $MnO_2/2rGO$ , or  $MnO_2/3GO$ based on the MnO<sub>2</sub>:rGO ratio, which is  $1:0.05$ ,  $1:0.1$ , or  $1:0.15$ , respectively.

To test the capacitive characteristics, the produced MnO2/rGO composites were used as an electrode. To begin, a mortar was prepared by adding a mixture of 8 parts water, 1 part super P, and 1 part PTFE to a mixture of 8 parts  $MnO<sub>2</sub>/rGO$ composite. A homogenous slurry was then created using DI water. Isopropanol and DI water were used to clean the Ni foam after it had been acid-treated to eliminate oxide residues before the slurry was cast. After cleaning the Ni foam, the resulting slurry was applied using a glass rod in a rolling procedure. The coated Ni foam was then placed in a laboratory oven set at 80C for 20 minutes. The two-electrode electrochemical system was tested in a 3M KOH electrolyte using cyclic voltammetry (CV) analysis with a scan rate ranging from 15 to 200  $mVs^{-1}$ . The synthesized  $MnO_2/rGO$  served as the working electrode, while a Pt plate served as the counter electrode. The potentiostat used for the analysis was a VersaSTAT4 from AMETEK, Inc. Analysis of the CV data allowed for the estimation of the individual capacitances and other factors.

#### **3.RESULTS AND DISCUSSION**

Composites containing  $MnO<sub>2</sub>$  and rGO sheets exhibit the presence of  $MnO<sub>2</sub>$  particles, indicating that the  $MnO<sub>2</sub>$  has been bonded to the rGO sheets. The rGO sheets are coated with a homogeneous distribution of MnO<sub>2</sub> particles when the ratio of MnO<sup>2</sup> to GO is 1:0.1. Raising the rGO concentration results in an uneven distribution of  $MnO<sub>2</sub>$  particles on the surfaces of the rGO sheets. Due to the huge aggregates produced by the increased rGO concentration, the surface of the  $MnO<sub>2</sub>/rGO$ composite is not homogeneous. Because of this, it is possible that the electrochemical process will benefit from the MnO2/2rGO composite's homogenous particle distribution. The XRD and Raman measurements are used to assess the structural properties and crystalline phases of the produced manganese oxide and MnO2/rGO composites. XRD patterns of different composites that were produced are shown in Figure 1a. Anatase phases are inferred to be present in the produced materials due to the diffraction peaks observed at 26.4 (101), 38.8 (004), 49 (200), 55 (105), 63.8 (204), 70.7 (220), and 76.2 (215) of MnO2. A good index to JCPDS No.: 21-1272 is seen in the obtained diffraction patterns. The crystallite size (D) of the specimen was evaluated through the Debye-Scherrer's equation. The predicted size of the  $MnO<sub>2</sub>$  crystallite is 14.8 nm. A significant reduction in crystallite size has been noted with the introduction of the GO. A crystallite size of around 13.7 nm was observed in the produced  $MnO<sub>2</sub>/2rGO$ . So, the rGO peak in MnO2/rGO composites is seen in Figure 1a. Nevertheless, additional diffraction peaks resemble  $MnO<sub>2</sub>$ , so validating the connection between rGO and MnO<sub>2</sub>. An optimum combination of rGO and  $MnO<sub>2</sub>$  materials would be  $MnO<sub>2</sub>/2rGO$  composite, since its rGO peak is more intense than that of other  $MnO<sub>2</sub>/rGO$ composites. Figure 1b shows the results of using Raman spectroscopy to study the  $MnO<sub>2</sub>/rGO$  composites' structural characteristics. The B1g, A1g, and Eg phonons are identified by the Raman bands at 435, 554, and  $682 \text{ cm}^{-1}$  in the synthesized  $MnO<sub>2</sub>$ , respectively. These Raman bands' presence verifies that anatase  $MnO<sub>2</sub>$  has formed. When it comes to  $MnO<sub>2</sub>/rGO$ composites, you can see the Raman shift bands at 1395 and 1625 cm-1 , which are all characteristic of MnO2, as well as the rGO

Raman bands [21], [22]. Additional evidence that a composite of MnO2 and rGO has formed is the presence of rGO peaks alongside MnO2 peaks.



**Figure 1.** (a) Raman shift spectrum and (b) XRD patterns of MnO<sub>2</sub> and MnO<sub>2</sub>/rGO composites.

For the MnO2/rGO composite, the XPS results are displayed in Figure 2 a-d. Figure 2a displays the XPS survey profile, which shows the binding energies of O 1s, Mn 2p, and C 1s. According to a high-resolution Ti2p XPS spectrum (Figure 2b), the binding energies at 464.9 eV and 459.1 eV, respectively, indicate Mn 2p1/2 and Mn 2p3/2. In addition, the reported MnO2 materials are in good agreement with the doublet Mn 2p binding energy, which shows a peak separation of 6.1 eV[23]. Figure 2c displays the O 1s XPS spectra after deconvolution. The Ti-O bond is compatible with the main core binding energy, which is 529.9 eV[24]. Alternatively, the CO and COO groups of the rGO moiety are responsible for the two binding energies of 530.7 and 531.6 eV, respectively. Furthermore, the deconvoluted C 1s XPS spectra exhibit three distinct binding energies at 285, 286.1, and 287.7 eV, as shown in Figure 2d. These energies correspond to the sp2 carbon, C O, and COO-groups in the MnO2/rGO composite[25], [26]. The binding energy of 283.5 eV, which may be due to the M C group, indicates that MnO2 interacts with the rGO molecule. Based on these findings, it's evident that the MnO2/rGO composite is quite effective, thanks to the excellent interaction between the two components.

Nitrogen (N2) adsorption-desorption isotherms were conducted to evaluate the surface areas and pore architectures of the produced  $MnO<sub>2</sub>/rGO$  composites. The results are displayed in Fig. 3a. Figure 3a displays the isotherm for the  $MnO<sub>2</sub>/2rGO composite, which has a specific surface area of$ 237.8 m2g -1 with holes sizes 2–8 nm. Regardless of the concentration of rGO in MnO2, the specific surface areas are decreased. In this case, the existence of micro and mesoporous surfaces in the

MnO2/2rGO composite, along with the favorable pore size distribution, suggests that these materials could be suitable for use as electrodes in capacitive devices. By conducting CV measurements in a 3 M KOH electrolyte at 298 K, the electrochemical capacitive properties of the electrochemical supercapacitors are assessed using the produced Manganese Oxide and MnO2/rGO composite electrodes as working electrodes. The CV plots of MnO2 (composite-1) and various MnO2/rGO (composite-2) composite electrodes are indicates in Figure 4a. The creation of electrochemical capacitance, often derived from faradaic pseudo-capacitance, is revealed by the low redox current response of the MnO2 electrode, which has a nonrectangular CV shape [27], [28]. The presence of rGO materials in composite-2 electrodes is indicated by a minor shift of the redox peaks towards the positive direction. The maximum redox current was observed in the composites-2 at 2rGO electrode, which indicates that this electrode is very effective when using the ideal ratio. Figure 4b displays the CV plots of the Manganese Oxide /2rGO composite electrode. To further understand the electrochemical behavior, the scan rate is different from 15 to 200 mVs-1 in a 3 M KOH electrolyte. Scan rates are positively correlated with the presence of marginal negative shifting, as seen in Figure 5 b. This finding is in line with the growing electrical polarization and irreversibility [29], [30]. By taking this into account, the process of producing capacitance or pseudo-capacitance by an irreversible Faradic reaction is clarified. The fact that CV plot shapes were affected by changes in scan rates provides more evidence that MnO2/rGO composite electrodes exhibit pseudo-capacitance behavior and electrochemical reversibility.





**Figure 2** (a) Profiling survey, (b) C 1s, (c) O 1s, and (d) Mn  $2pXPS$  spectrum of  $MnO<sub>2</sub>/2rGO$  composites



**Figure 3** (a) Distribution plots for pore size and (b) N<sub>2</sub> adsorption–desorption isotherm for  $MnO_2/2rGO$  composites



**Figure 4** (b) CV plots in 3M KOH electrolyte, and (a) A series of CV plots at various scan rates for MnO2/2rGO composite electrode

Figure 5 as shows the specific Cs of the Manganese Oxide and MnO2/2rGO composite electrodes, which are used to evaluate the capacitive characteristics. When compared to  $MnO<sub>2</sub>$ and other  $MnO_2/rGO$  composite electrodes, the  $MnO_2/2rGO$ electrode shows the highest Cs at  $15 \text{ mVs}^{-1}$  at  $343.7\text{Fg}^{-1}$ . The  $MnO<sub>2</sub>$  electrode has a very low Cs value of 76.5Fg<sup>-1</sup>, but when rGO is added to it, the Cs value increases greatly, showing that rGO is important in composites. When compared to a  $MnO<sub>2</sub>$ electrode, the Cs value of a composite electrode made of  $MnO<sub>2</sub>$ and GO is five times higher. The increase in electrical conductivity and morphological improvement caused by rGO are primarily responsible for this Cs enrichment. Thus, it may help improve the speed of ion transport and charge transfer. The retention of capacitance from the  $MnO<sub>2</sub>/rGO$  composite electrode increases from 15 to 200 mVs<sup>-1</sup>, a 33% increase over its initial value. The drop in capacitance at high scan rate can be because the material isn't as efficient. The results were confirmed by performing the galvanostatic charge discharge (GCD) test on the Manganese Oxide /rGO composite supercapacitor electrode. This study was conducted over a potential between 0 to 1.5 V using current densities ranging from 0.6 to 10 A/g. Figure 6a displays the outcomes of this analysis. A charge storage mechanism supported by pseudocapacitive nature is shown by the GCD curves, which exhibit a small divergence from the normal shape of symmetric triangles. This process probably occurs because of the electrolyte's K+ ions interacting with MnO2/rGO [31], [32]. As current densities increase, the specific capacitance value decreases because the charging and discharging durations get shorter. Supercapacitors made of MnO2/rGO composites showed specific capacitance values of 320.6, 169.2, 95.4, and 60.7 Cg-1 at 0.6, 1.2, 2.4, and 6 Ag-1 current densities as shown in Figure 6 b. Since ions may only penetrate the electrode material to a certain depth, the capacitance value drops as the current densities increases. The high Cs of 320.6 Cg-1 is a consequence of the prolonged discharge period and rapid ion diffusion, both of which are caused by the mixed porous structure of MnO2/rGO composites [33]. On the other hand, increasing the concentration of rGO in composites leads to a drop in capacitance value because electrochemical processes cause the creation of many aggregates [34], [35]. The charge transfer between electrode materials and at contact interfaces is greatly hindered as a result of the ions diffusion length being prolonged. As indicates in Fig 7, the electrochemical characteristics of the MnO2/rGO electrode composite are tested utilizing the multicycles CVs to further verify its cycling capabilities. With excellent capacitive performance, rate capability, and high cyclic ability, the composite-2 electrode retains more than 90% of the starting value even after 2000 cycles. Owing to the dual impacts of rGO on the composite, which increase electrical conductivity and surface-to-volume ratio, the composite-2 electrode in this study shows an excellent cycling capability in an alkaline electrolyte and a high Cs value. Consequently, it may promote rapid charge transfer and ion transit.



**Figure 5** (a) Specific capacitances of various electrodes, and (b) evaluation of specific capacitance and scanning rate of MnO2/2rGO composite electrode



**Figure 6** (a) Evaluation of Cs from GCD curvatures and current densities and (b) comparison of GCD curves with current densities for Manganese Oxide/2rGO composite electrode



Figure 7. The MnO<sub>2</sub>/2rGO composite electrode maintains its cycle stability even after 2000 cycles.

## **4. CONCLUSIONS**

In electrochemical supercapacitor production, it serves as an electrode, the effective  $MnO<sub>2</sub>/rGO$  composites are synthesized using a straightforward solution technique. Synthesizing  $MnO<sub>2</sub>$ using the sol-gel method is the first step in preparing  $MnO_2/rGO$ composites. Next, varying amounts of rGO are added to the produced MnO2 using the sonochemical approach. Verification of the composite's  $rGO$  and  $MnO<sub>2</sub>$  integration is confirmed by several characterizations. By taking CV readings, the capacitive characteristics of  $MnO<sub>2</sub>/rGO$  composite electrodes can be studied. One possible explanation for the higher Cs value of  $343.7\text{Fg}^{-1}$  at 15 mVs<sup>-1</sup> achieved by the Manganese Oxide /rGO composite electrode compared to  $MnO<sub>2</sub>$  and other electrodes is that the addition of rGO to the electrodes increases the electrical

conductivity and the surface-to-volume ratio. Superior cycling performance is demonstrated after 2000 cycles by the MnO2/rGO composite electrode. Hence, the results show that electrode materials improved electrochemical behavior can be MnO2/rGO composites with strong capacitive performance.

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